



Defects in Materials

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Cray Users Group Meeting,
May 2016.



- Introduction.
 - Motivation.
 - Methods for calculating material properties.
- Defects in materials.
 - Why are defects challenging?
 - Oxygen vacancies in α -Alumina.
- Conclusions.



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Why α -alumina?



Why α -alumina?



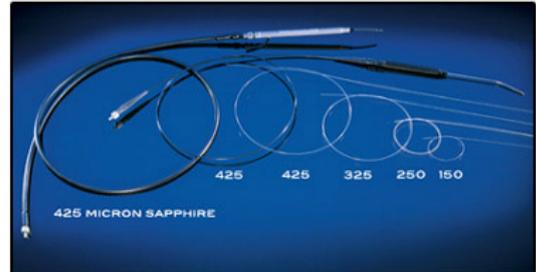
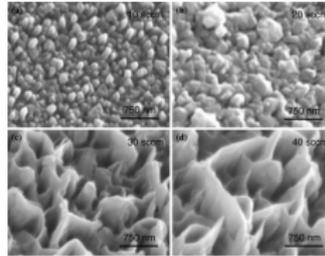
- High-temperature structural ceramics

Why α -alumina?



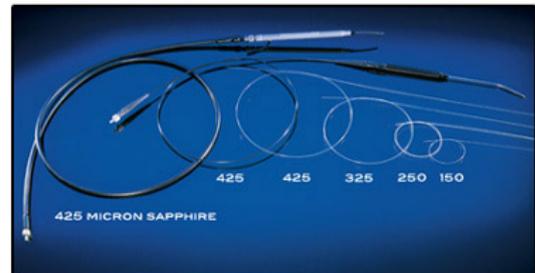
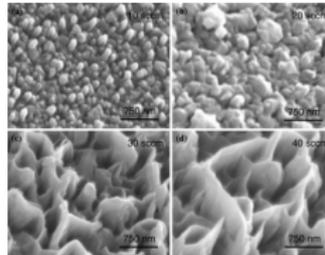
- High-temperature structural ceramics
- Optical devices

Why α -alumina?



- High-temperature structural ceramics
- Optical devices
- Semiconductor manufacturing

Why α -alumina?



- High-temperature structural ceramics
- Optical devices
- Semiconductor manufacturing
- Mechanical usage

Importance of studying defects



- Defects are ubiquitous in materials and thermodynamically unavoidable at finite temperature.

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- Strongly dominate a wide variety of material properties i.e, optical, mechanical, electrical, and other transport properties.

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Importance of studying defects



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- Strongly dominate a wide variety of material properties i.e, optical, mechanical, electrical, and other transport properties.
- The performance and long-term stability of devices is often governed by the creation, transport and annihilation of point defects.
- Product manufacturing process can be improved by studying the defect formation at different environmental conditions.



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Methods – Ground-state properties



Properties that are intrinsic to a system with all its electrons in equilibrium.

Methods – Ground-state properties



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- Density functional theory is the “standard model” for understanding **ground-state** properties.
- Total energy is a functional of the charge density.
- Kohn-Sham formulation: Map the interacting many-electron problem to non-interacting electrons moving in a self-consistent field.

$$\left(-\frac{\nabla^2}{2} + V_{\text{ionic}}(\mathbf{r}) + V_{\text{Hartree}}(\mathbf{r}) + V_{\text{xc}}(\mathbf{r}) \right) \psi(\mathbf{r}) = \epsilon \psi(\mathbf{r})$$

Methods – Ground-state properties



Properties that are intrinsic to a system with all its electrons in equilibrium.

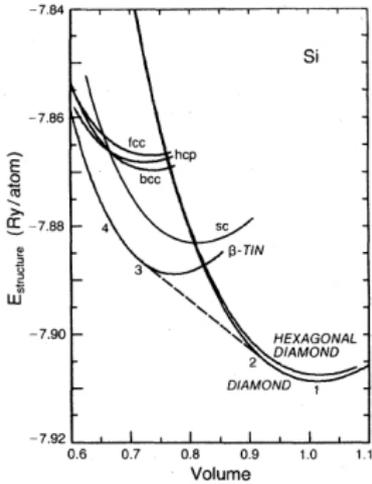


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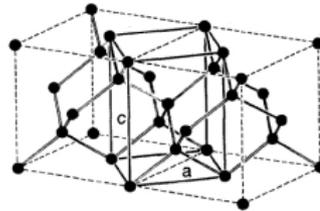
$$\left(-\frac{\nabla^2}{2} + V_{\text{ionic}}(\mathbf{r}) + V_{\text{Hartree}}(\mathbf{r}) + V_{\text{xc}}(\mathbf{r}) \right) \psi(\mathbf{r}) = \epsilon \psi(\mathbf{r})$$

Local density approximation
Generalized gradient approximation

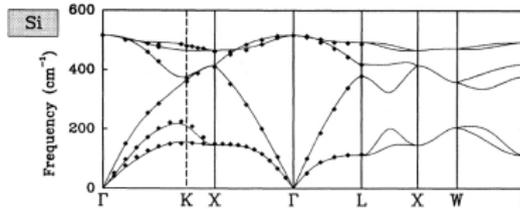
Methods – Ground-state properties



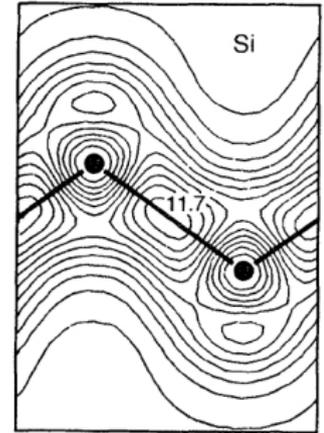
Phase transitions¹



Crystal Structure¹



Phonons²



Charge density¹

¹M. T. Yin and M. L. Cohen, Phys. Rev. B **26**, 5668 (1982).

²P. Giannozzi, S. de Gironcoli, P. Pavone, and S. Baroni, Phys. Rev. B **43**, 7231 (1991).

Methods – Excited-state properties



Spectroscopic properties that involve experiments creating an excited particle above the ground state.

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- Concept and formalism of **interacting particle Green's function** (G).

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- Many-body perturbation theory is the “standard model” for understanding **excited-state** properties.

$$G^{-1} = G_0^{-1} - \Sigma$$

Methods – Excited-state properties



Spectroscopic properties that involve experiments creating an excited particle above the ground state.

- Concept and formalism of **interacting particle Green's function** (G).
- Many-body perturbation theory is the “standard model” for understanding **excited-state** properties.

$$G^{-1} = G_{\text{DFT}}^{-1} + \Sigma - V_{\text{xc}}$$

- GW approximation to the self-energy (Σ).

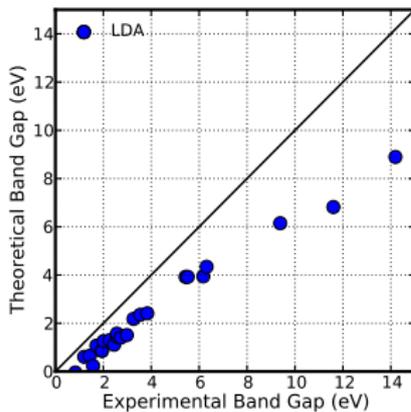
$$\Sigma = \text{---} \bullet \text{---} \overset{\curvearrowright}{\text{---}} \bullet \text{---} = iGW$$

$$W = \varepsilon^{-1} v$$

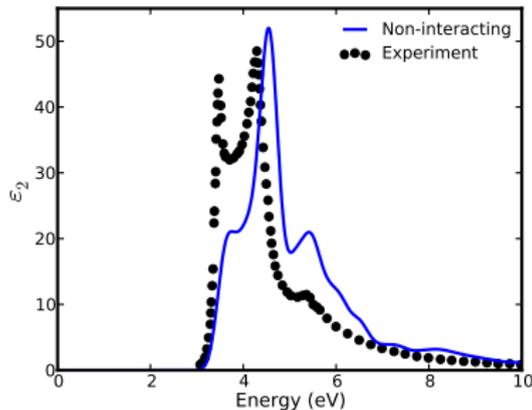
Methods – Excited-state properties



Materials: InSb,
InAs, Ge,
GaSb, Si, InP,
GaAs, CdS,
AlSb, AlAs,
CdSe, CdTe,
BP, SiC, C₆₀,
GaP, AlP,
ZnTe, ZnSe,
c-GaN, w-GaN,
InS, w-BN,
c-BN,
diamond, w-AlN,
LiCl,
Fluorite, LiF



Quasiparticle Gap¹



Optical absorption²

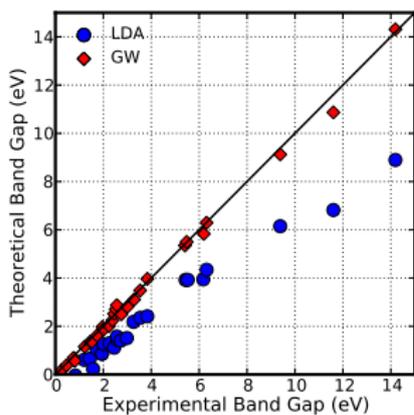
¹S. G. Louie in Topics in Computational Materials Science, edited by C. Y. Fong (World Scientific, Singapore, 1997).

²J. Deslippe, G. Samsonidze, D. A. Strubbe, M. Jain, M. L. Cohen, and S. G. Louie, Comput. Phys. Commun. **183**, 1269 (2012).

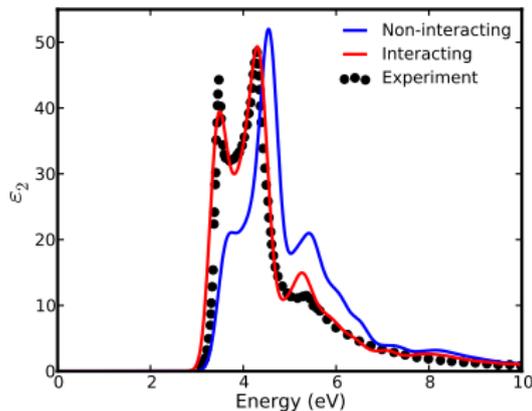
Methods – Excited-state properties



Materials: InSb,
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GaSb, Si, InP,
GaAs, CdS,
AlSb, AlAs,
CdSe, CdTe,
BP, SiC, C₆₀,
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ZnTe, ZnSe,
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diamond, w-AlN,
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Fluorite, LiF



Quasiparticle Gap¹



Optical absorption²

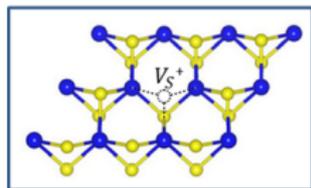
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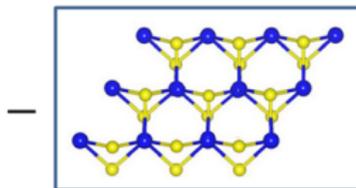
Formation energy and charge transition level



$$E_q^f[\vec{R}_q](E_F) = E_q[\vec{R}_q] - E_{\text{ref}} + \mu_S + q(E_F + E_V)$$

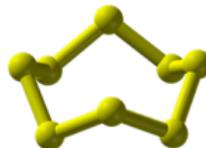


MoS₂ with mono-vacancy of sulfur, V_S^+

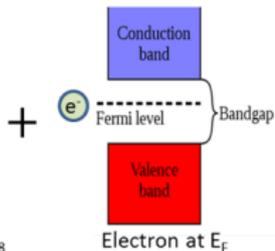


Pristine MoS₂

$$+ \frac{1}{8}$$



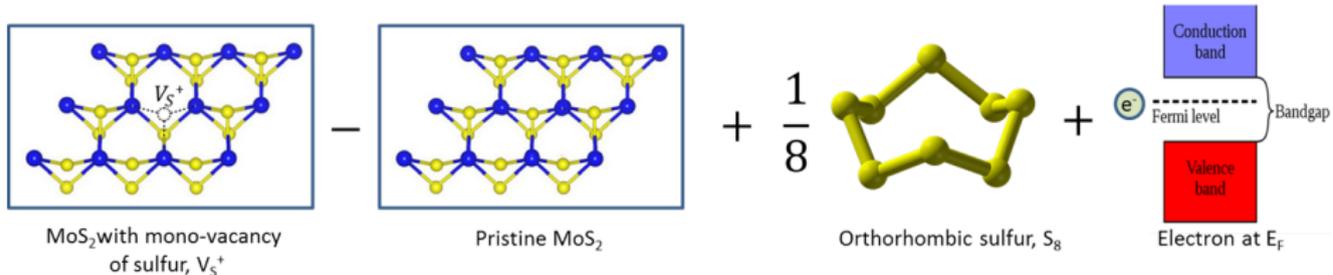
Orthorhombic sulfur, S₈



Formation energy and charge transition level



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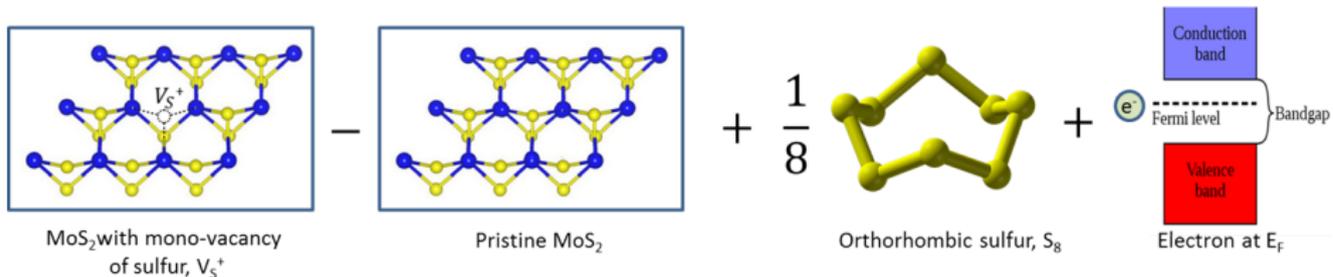


Charge transition level : $\varepsilon^{q/q-1} = \text{Fermi energy where defect } q \rightarrow q - 1.$
 $= E_{q-1}^f[\vec{R}_{q-1}](E_F = 0) - E_q^f[\vec{R}_q](E_F = 0)$

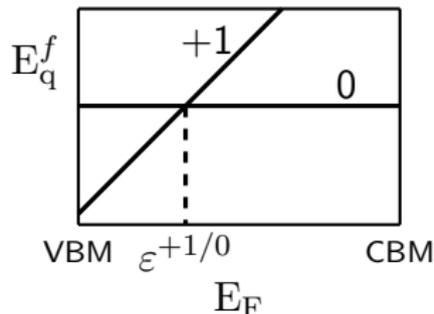
Formation energy and charge transition level



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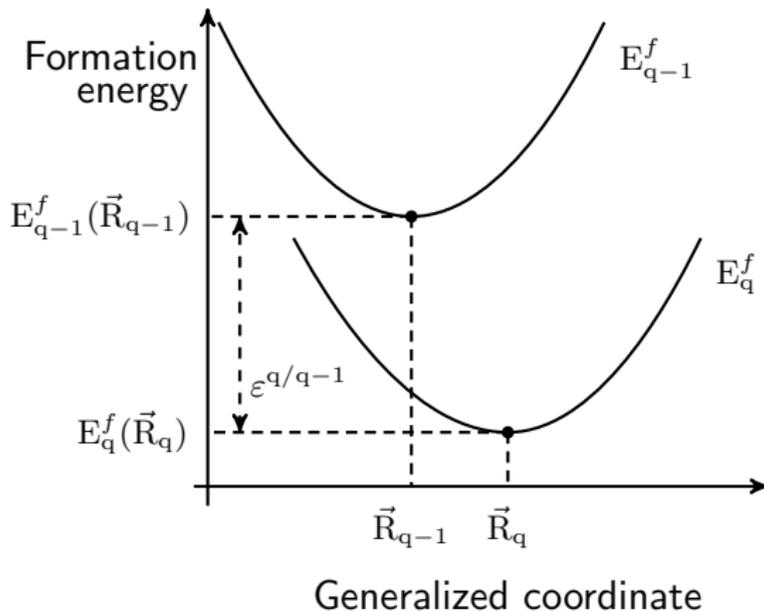
Charge transition level : $\varepsilon^{q/q-1} = E_F$ where defect $q \rightarrow q - 1$.

$$= E_{q-1}^f[\vec{R}_{q-1}](E_F = 0) - E_q^f[\vec{R}_q](E_F = 0)$$


Formation energy and charge transition level



DFT + GW methodology.



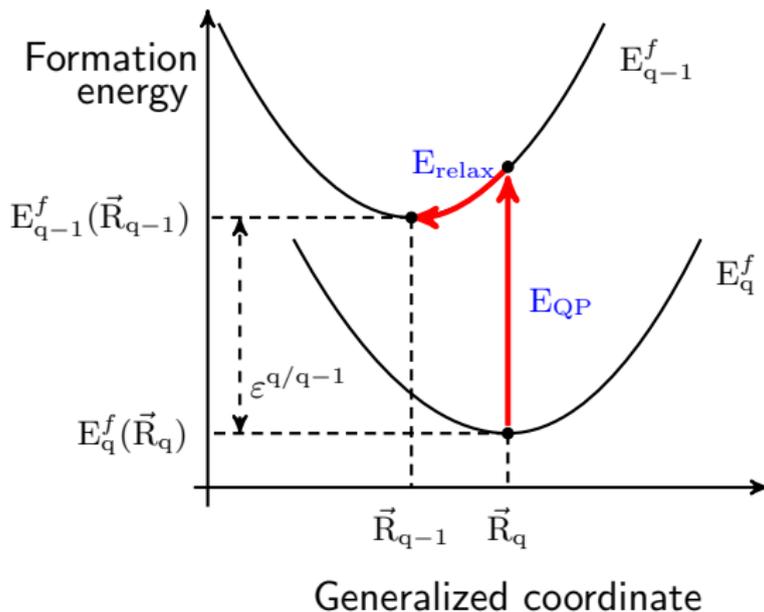
M. Jain, J. R. Chelikowsky and S. G. Louie, Phys. Rev. Lett. **107**, 216803 (2011).

A. Malashevich, M. Jain and S. G. Louie, Phys. Rev. B **89**, 075205 (2014).

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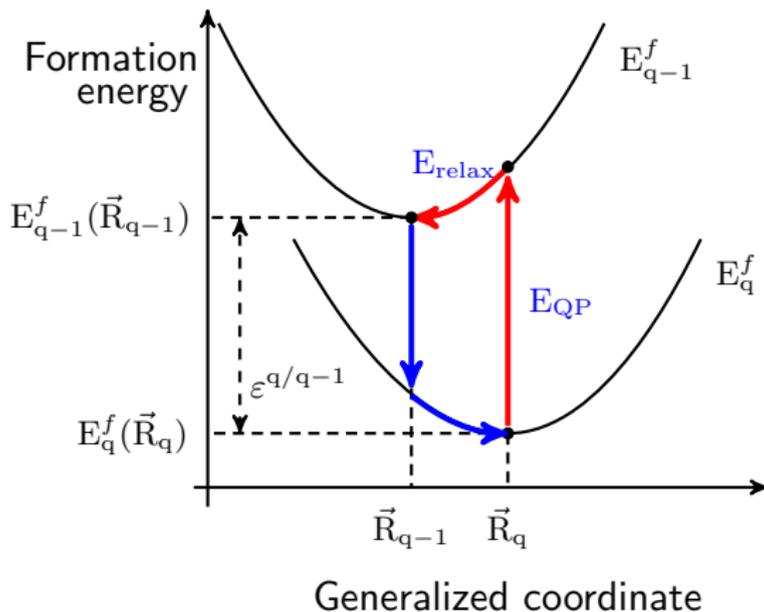
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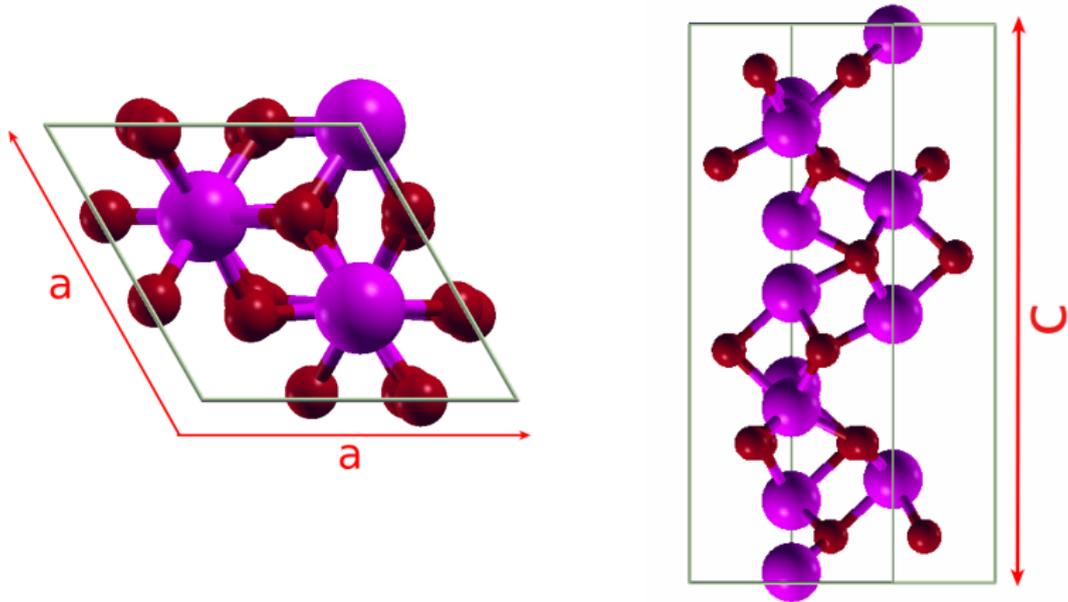


- Potentially strong electron-electron correlations.
 - Can be open-shell systems.
 - Multiple localized, interacting electrons.
- Lattice relaxation effects.
- Screening from the host.
 - Mimicking the system by isolated cluster may be incorrect.
- Experiments often involve excited-state properties (deep level transient spectroscopy or optical absorption etc.)
- Computational difficulty – isolated defect.



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Structure of α -alumina



Hexagonal unit cell (space group $R\bar{3}C$) contains 30 atoms



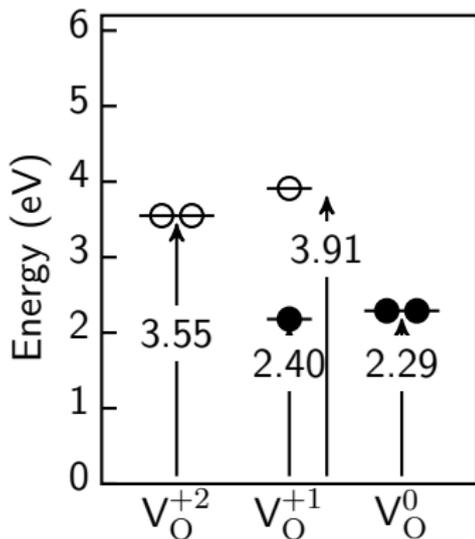
DFT calculation

- Quantum Espresso package
- Norm-conserving Pseudopotentials
- PBE exchange correlation
- Wavefunction cutoff: 75 Ry
- 2x2x2 kgrid for 120 atom supercell (2x2x1)
- Only Γ point for 270 atom supercell (3x3x1)

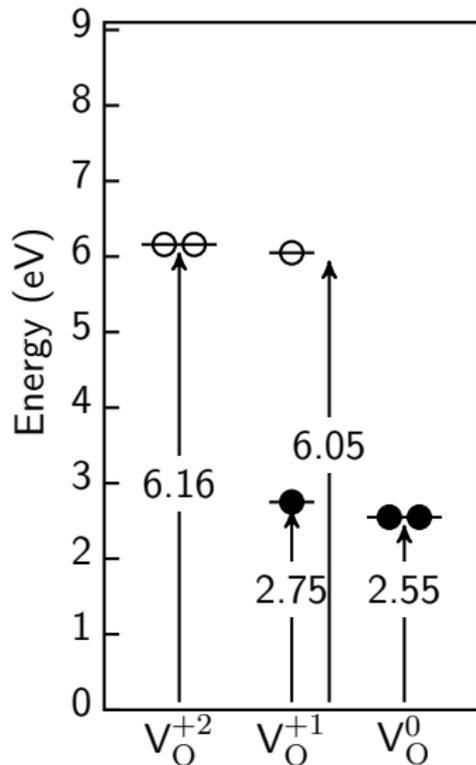
GW calculation

- BerkeleyGW package
- 25 Ry cutoff for static dielectric matrix
- For finite frequency used GPP (Generalized Plasmon Pole)
- 4000 bands for 120 atom supercell
- 9000 bands for 270 atom supercell

PBE and quasiparticle level diagram - V_O



PBE

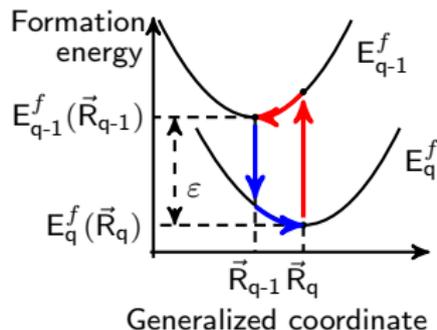


GW

Results: Charge transition levels



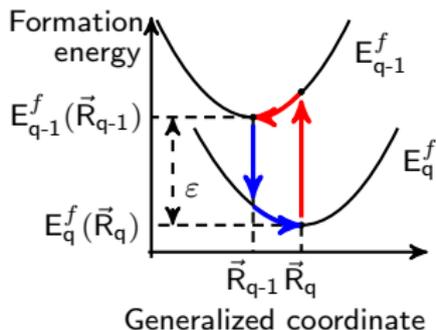
CT levels	P1	P2	Δ
$\epsilon_{120}^{+1/0}$	4.95	3.65	1.30
$\epsilon_{270}^{+1/0}$	4.87	3.79	1.08
$\epsilon_{120}^{+2/+1}$	5.21	3.79	1.42
$\epsilon_{270}^{+2/+1}$	5.05	3.85	1.20



Results: Charge transition levels



	without correction			with correction			
CT levels	P1	P2	Δ	P1	P2	Δ	Mean
$\epsilon_{120}^{+1/0}$	4.95	3.65	1.30	3.77	3.65	0.12	3.71
$\epsilon_{270}^{+1/0}$	4.87	3.79	1.08	3.93	3.79	0.14	3.86
$\epsilon_{120}^{+2/+1}$	5.21	3.79	1.42	2.86	2.62	0.24	2.74
$\epsilon_{270}^{+2/+1}$	5.05	3.85	1.20	3.17	2.91	0.26	3.04



Results: Comparison with literature



CT levels	DFT+GW	HSE ¹	GGA ²	GGA
$\epsilon^{+1/0}$	3.8 ± 0.1	4.1	5.2	5.65
$\epsilon^{+2/+1}$	2.9 ± 0.1	3.2	5.0	4.35

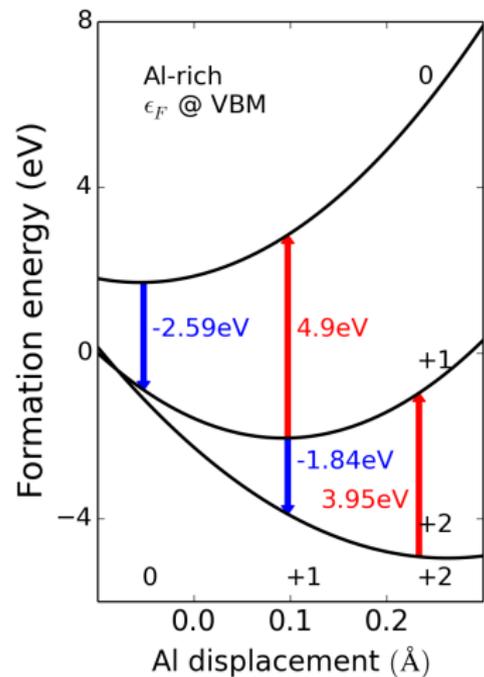
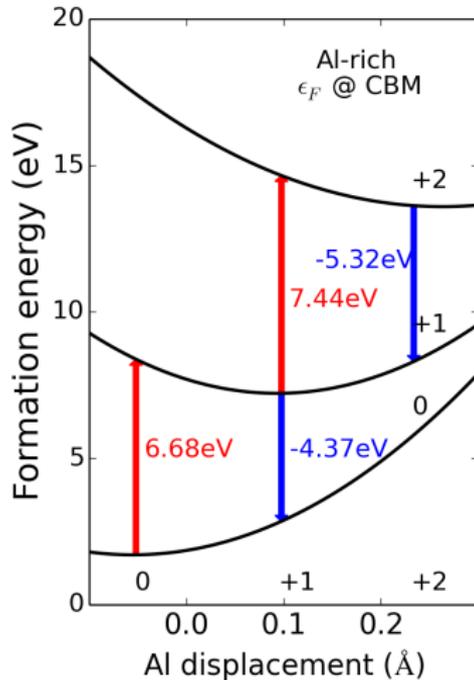
- Ref. [1] HSE calculation with a band gap of 9.2 eV.
- Ref. [2] GGA with a scissors shift.

¹M. Choi, A. Janotti and C. G. van de Walle, J. Appl. Phys. **113**, 44501 (2013).

²K. Matsunaga et al, Phys. Rev. B **68**, 85110 (2003).



Configuration coordinate diagram



	V_O^0	V_O^{+1}	V_O^{+2}
Al (Nearest Neighbors)	-0.052	0.098	0.234
O (Next Nearest Neighbors)	-0.011	-0.034	-0.067

Comparison to experiments

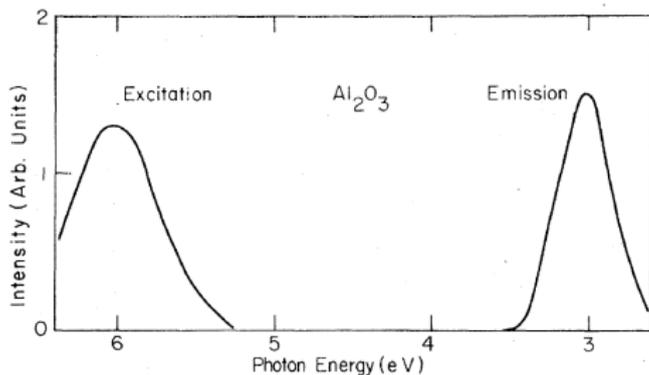
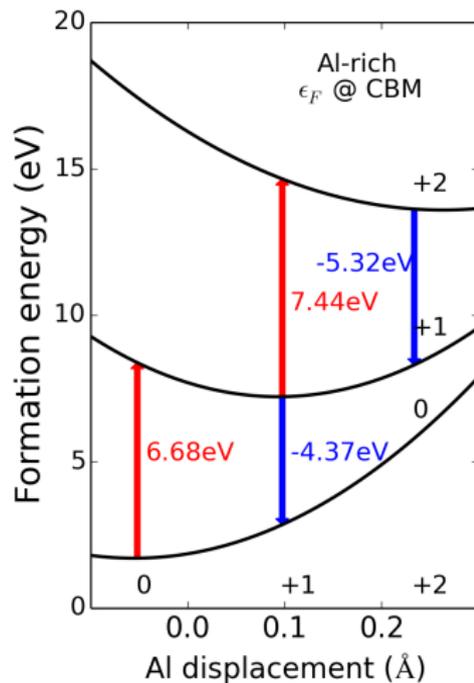


FIG. 2. Excitation and emission spectra of the F center from subtractively colored Al_2O_3 measured at 300 K.



K. H. Lee and J. H. Crawford, Phys. Rev. B **19**, 3217 (1979).



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- DFT and GW are powerful techniques to calculate ground-state and excited-state properties of materials from first principles.
- DFT+GW method combines these to calculate properties of defects from first principles.
- Used DFT+GW to understand and characterize F centers in α -alumina.

Acknowledgements



Tathagata Biswas



Mit H. Naik

Computations done at SERC on SahasraT!